

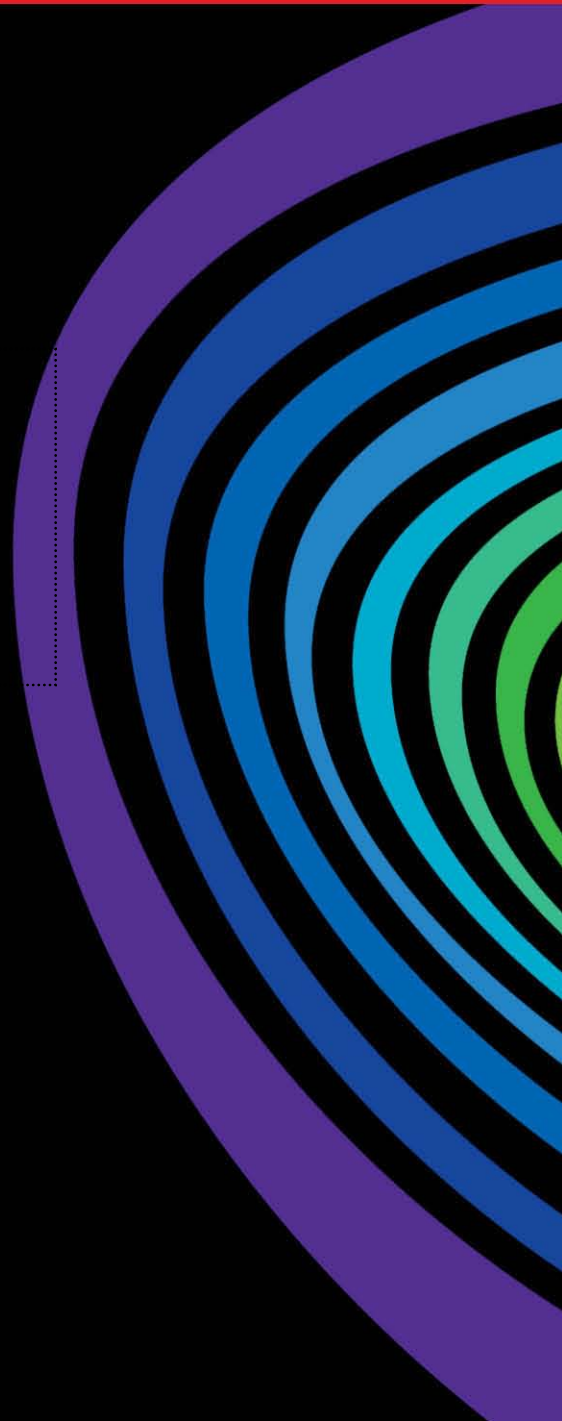
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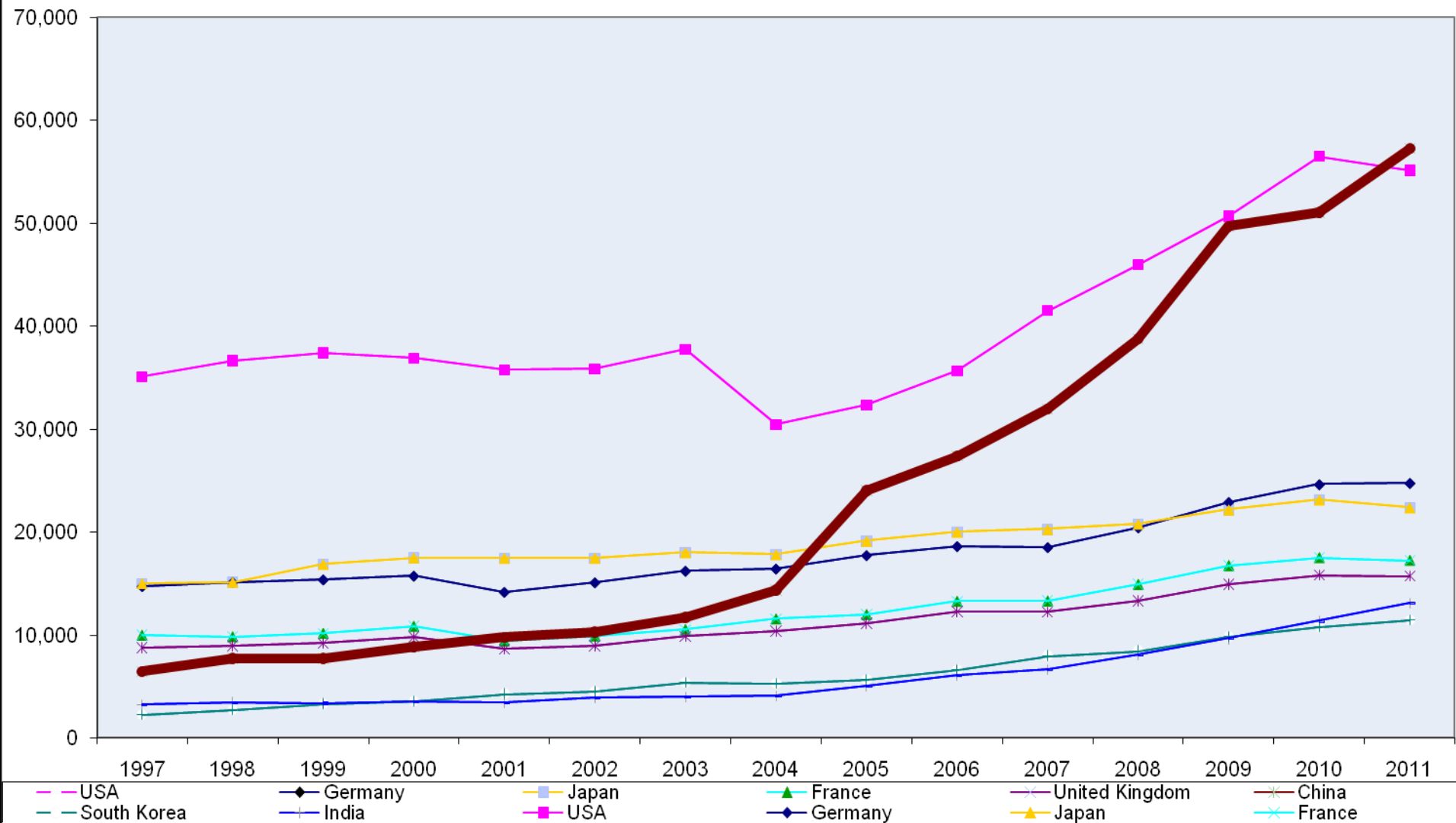


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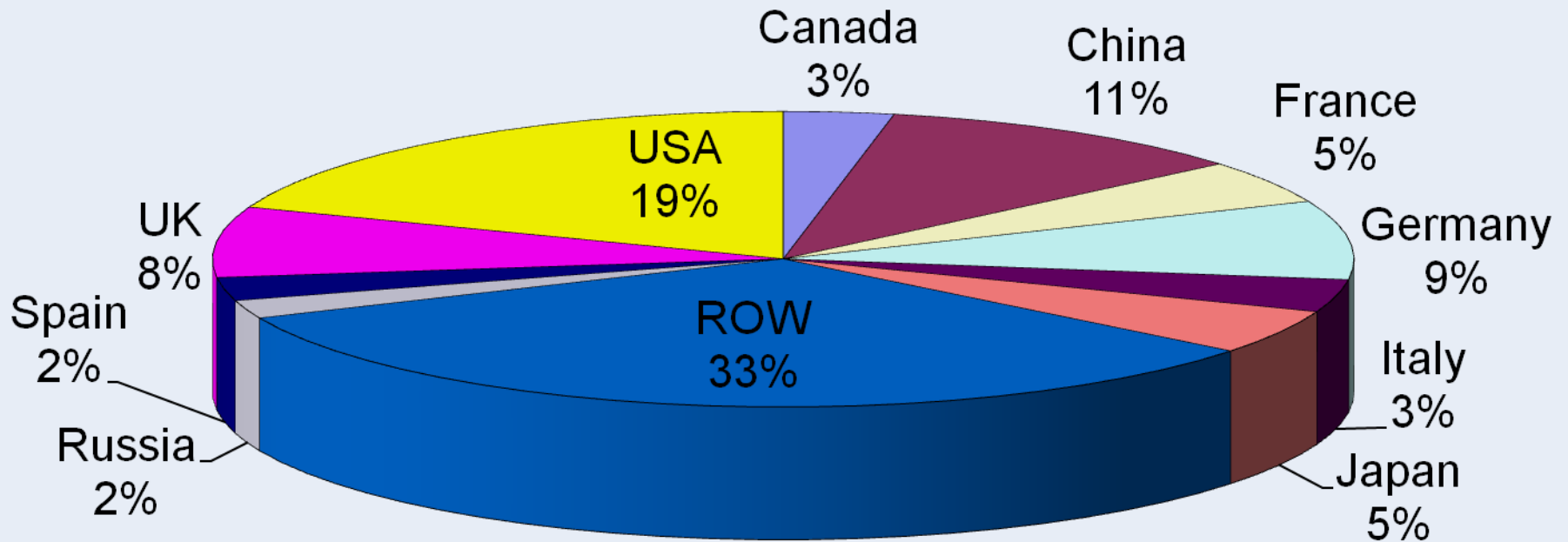


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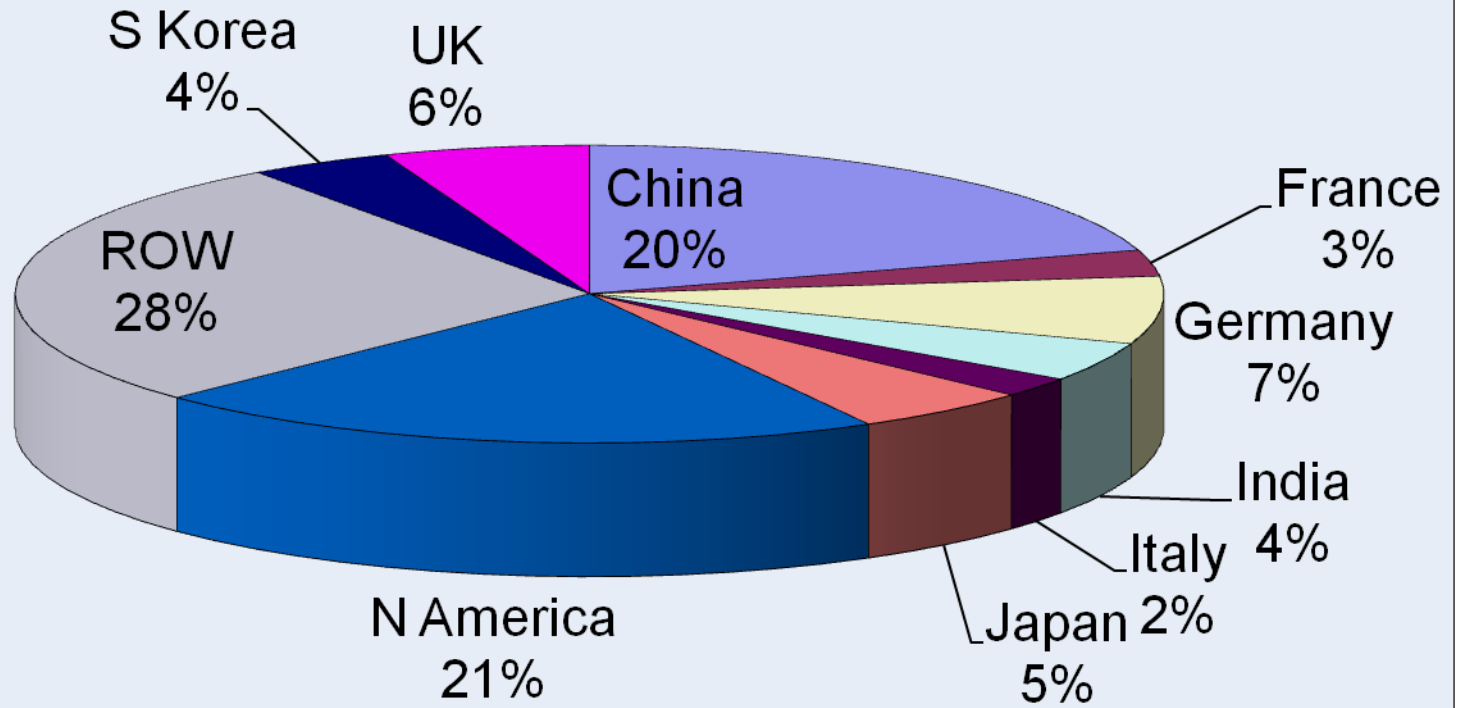
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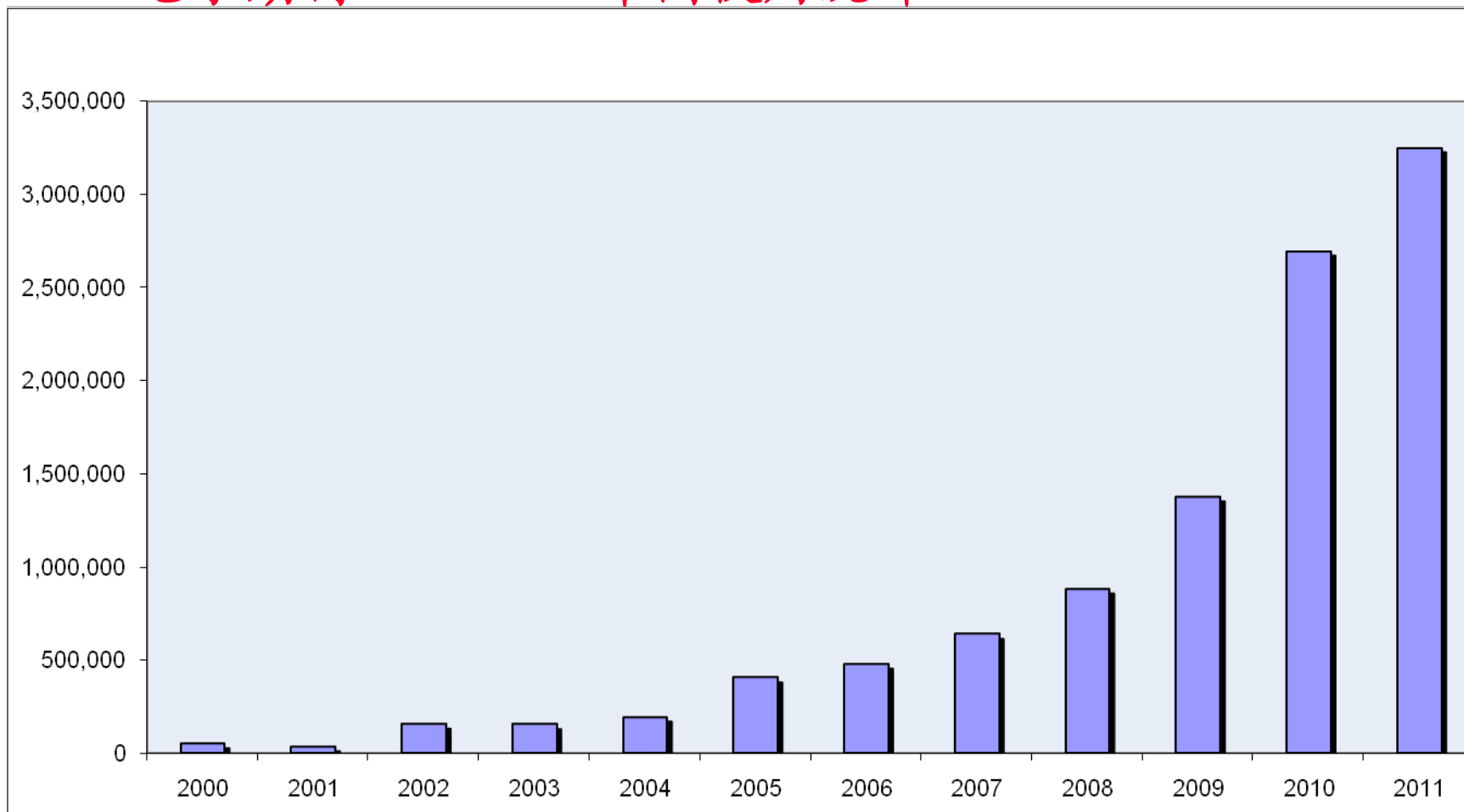


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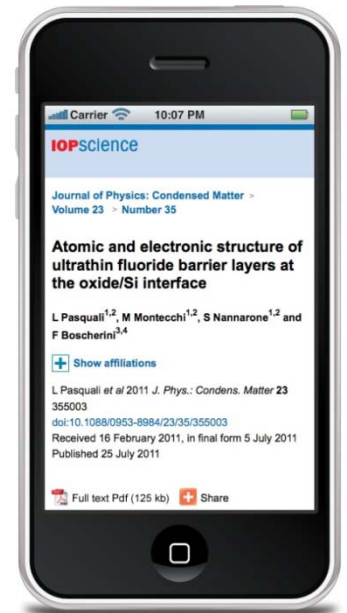


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From the intensity ratio of the interface and bulk Si 2p components, $S_{I_{int}}$ and $S_{I_{bulk}}$, it is possible to evaluate the thickness of the intermixed region [36]. The intensities of the two components are obtained from the fitting of the experimental spectra with Voigt peaks (inset of figure 1 (a)), centred at 99.5 eV (Bulk-FWHM 1.5 eV) and at 102.5 eV (Int.-FWHM 2.0 eV). Assuming that the buried reacted interface is uniform and using the calculated inelastic mean free path (IMFP) values [26] of Si and SiO₂ for the substrate and the interface layer, respectively, the thickness of this intermixed region is estimated to be about 9 Å. Applying calculated IMFPs for a Yb silicate interface layer does not alter this estimate appreciably. It should be noted that, in general, thicker interface reacted layers (1–5 nm) were suggested for Yb₂O₃ films prepared by the ALD method [11] and by reactive sputtering [32, 33].

Figure 5. 2D He diffraction patterns spanning (a) growth regime IV (0.72 ML coverage) and (b) growth regime V (1.75 ML coverage), both collected at $\theta_i = 58.2^\circ$. The color map is logarithmic. The unit cells, corresponding to the two possible surface domains of $(3 \times 3)R12.5^\circ$ symmetry, are represented by the solid lines in (a). In (b), the lower portion is experimental data, whilst the upper portion derives from a simulation, as discussed in the text.

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Krastanov growth mode for SiF₂ can thus be assumed. It should be noted however, that due to high rate of molecular re-evaporation from the surface when this is held at 700–750°C during growth, the density of the SiF₂ islands does not increase rapidly, even after prolonged deposition times. The surface continues to expose large portions of the uncovered, reacted wetting layer.

Data presented in figure 2 correspond to a single layer of SiF₂ deposited on clean Si held at 700°C. The F 1s and Sr 3d photoemission signals shown in figures 2(a) and (b) indicate the formation of the wetting layer. It can be noticed that the peak positions on the interface layer are shifted to lower BE with respect to the bulk SiF₂ values (also shown for comparison in figures 2(a) and (b)). In particular, the BE of the F 1s core level is shifted from 685.9 ± 0.2 (bulk) to 684 ± 0.2 eV (int.), as for the case of CaF₂/Si(100) [16]. Concerning Sr 3d, a unique doublet corresponding to the 1.8 eV $3d_{5/2} - 3d_{3/2}$ spin-orbit split components is observed for the interface layer, with maximum of the Sr $3d_{5/2}$ peak at 134.5 ± 0.2 eV. This is shifted by about 1.5 eV with respect to the bulk level and it is ascribed both to initial and final state effects in the photoemission process [19]. The observed shifts can be considered as the evidence of the formation of the reacted wetting layer, with the dissociation of the SiF₂ molecules and with the Sr atoms bonding to the substrate Si to form Si–Sr–F species.

a) 600 | — Si/SrF₂ int. b) 300 | — Si/SrF₂ int.

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N R Wilson *et al* 2010 *New J. Phys.* **12** 125010 doi: 10.1088/1367-2630/12/12/125010

On the structure and topography of free-standing chemically modified graphene

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N R Wilson^{1,4}, P A Pandey¹, R Beanland¹, J P Rourke², U Lupo¹, G Rowlands¹ and R A Römer^{1,3}

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Part of Focus on Chemically Modified Graphene

The mechanical, electrical and chemical properties of chemically modified graphene (CMG) are intrinsically linked to its structure. Here, we report on our study of the topographic structure of free-standing CMG using atomic force microscopy (AFM) and electron diffraction. We find that, unlike graphene, suspended sheets of CMG are corrugated and distorted on nanometre length scales. AFM reveals not only long-range (100 nm) distortions induced by the support, as previously observed for graphene, but also short-range corrugations with length scales down to the resolution limit of 10 nm. These corrugations are static not dynamic, and are significantly diminished on CMG supported on atomically smooth substrates. Evidence for even shorter-range distortions, down to a few nanometres or less, is found by electron diffraction of suspended CMG. Comparison of the experimental data with simulations reveals that the mean atomic displacement from the nominal lattice position is of order 10% of the carbon-carbon bond length. Taken together, these results suggest a complex structure for CMG where heterogeneous functionalization creates local strain and distortion.



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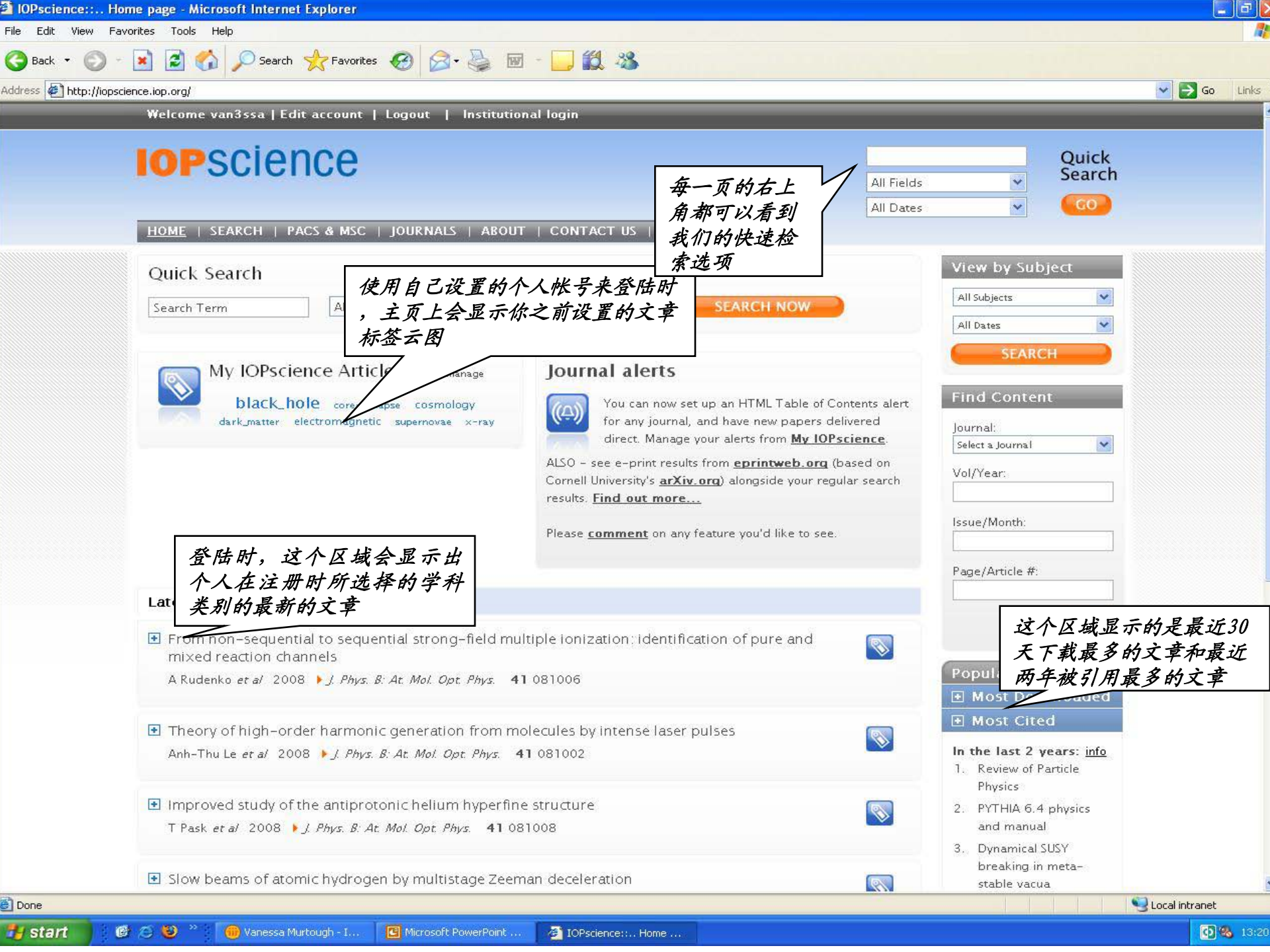
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Spherical symmetry in $f(R)$ -gravity

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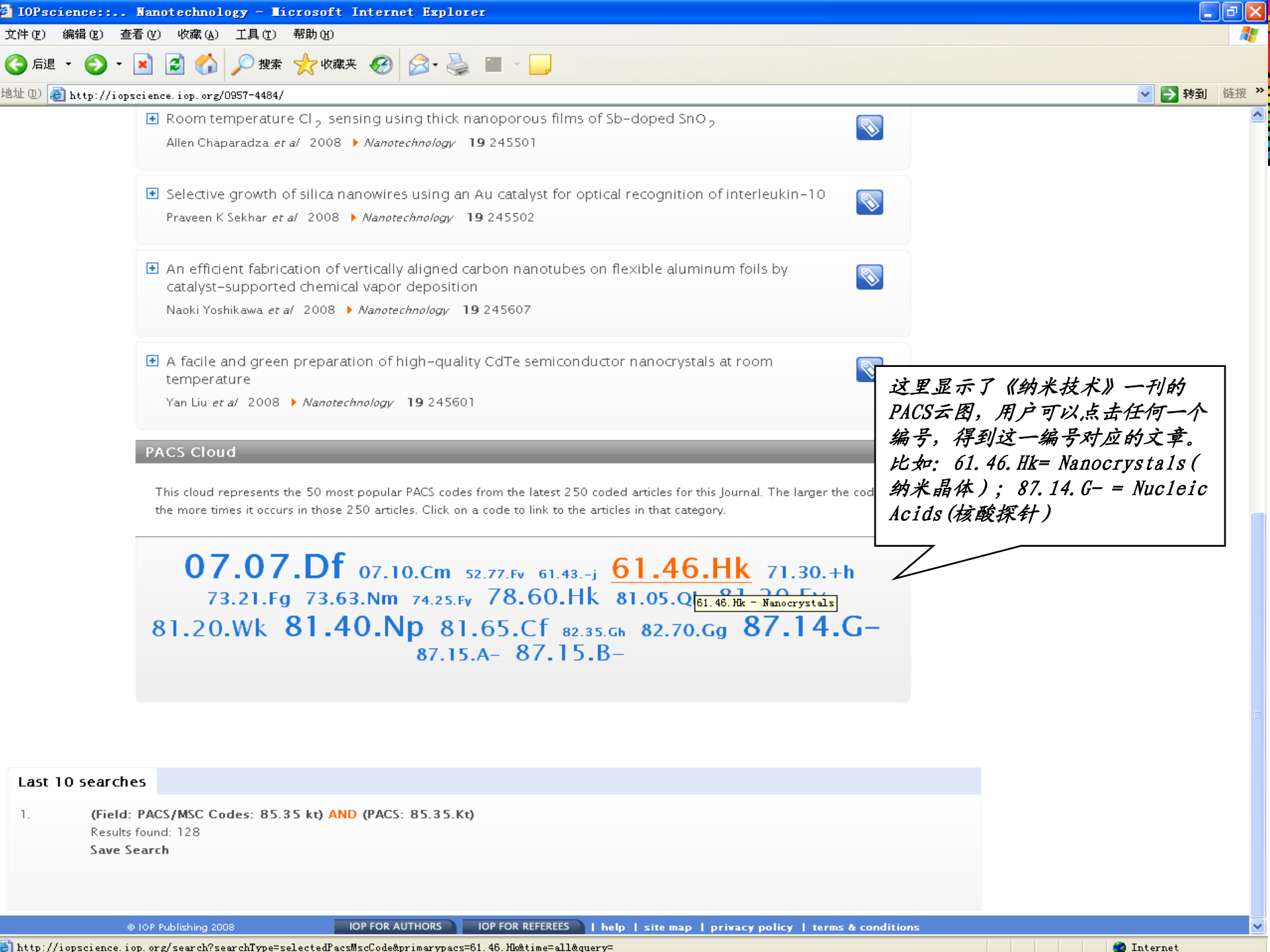
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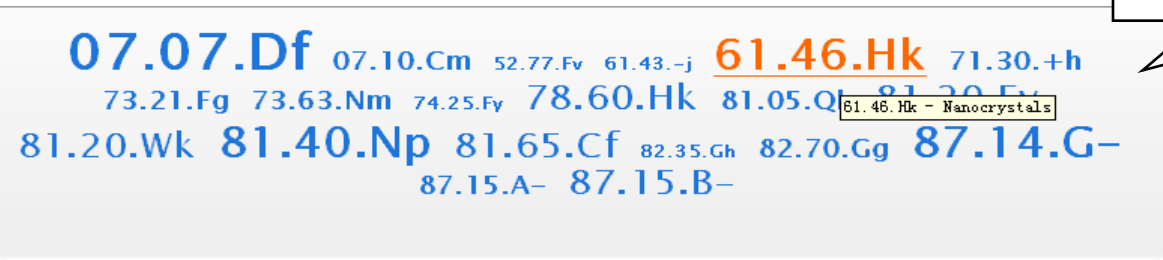
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